

**Exercise 1 Density of dislocations and the Frank lattice**

The dislocation density in a crystal is defined as the ratio between the total dislocation length and the crystal volume. The dislocations arriving at the surface of the crystal create local deformation (etch-pits) that we can observe. Knowing the average distance between each etch-pit, i.e., plastic deformation is uniform, we can calculate the density of the dislocations in the crystal.

In general, etch-pits are produced by screw dislocations. We observe the surface of an aluminum sample ( $1 \text{ cm}^3$ ) perpendicularly to the direction  $[\bar{1}10]$ , and we measure an average distance between etch-pits of  $10 \mu\text{m}$ .

Calculate the maximum plastic deformation due to the movement of dislocations with Burgers vector  $b = \frac{b}{2}[\bar{1}10]$  in the dense planes relative to this vector. We neglect dislocation sources. The lattice parameter of silicon is  $5.43\text{\AA}$ , and the atomic radius is  $1.11\text{\AA}$ .

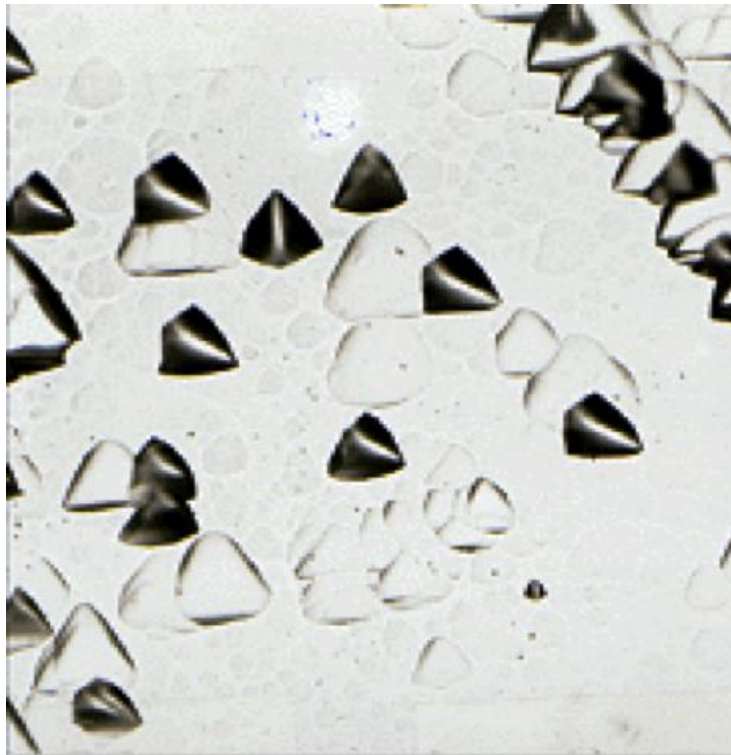


Figure 6.1 Etch-pits in silicon observed on a (111) plane

## Exercise 2 Nabarro-Herring creep

A cubic sample of dimension  $d$  is submitted to a constant tensile stress  $\sigma$ . Calculate the strain rate of the sample maintained at a constant temperature  $T$  due to the diffusion of the atoms (respectively of the vacancies) within the crystal.

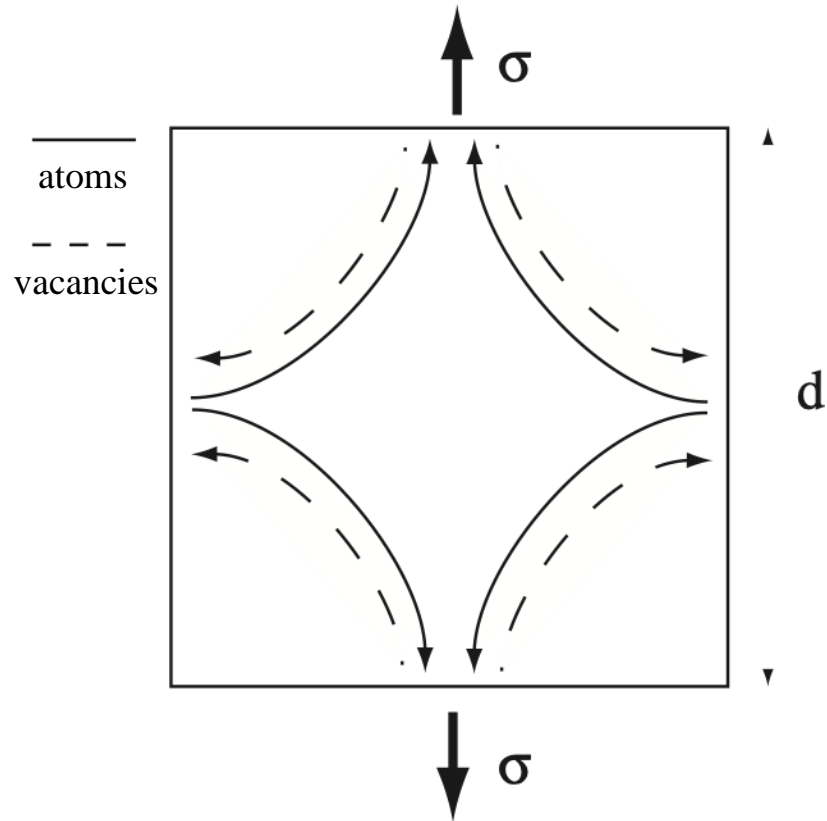


Figure 6.2 Scheme of the vacancy and atoms movement during creep